Surface and Interface Free-Carrier Depletion in GaAs Molecular Beam Epitaxial Layers: Demonstration of High Interface Charge

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Surface and interface free-carrier depletion in GaAs molecular beam epitaxial layers: Demonstration of high interface charge

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Molecular beam epitaxial GaAs layers of electron concentration \(1.69 \times 10^{17} \text{ cm}^{-3}\), and various thicknesses \(d = 0.25, 0.50, 1.00, \text{ and } 2.00 \mu \text{m}\), have been grown on semi-insulating GaAs substrates and characterized by the Hall effect and capacitance-voltage \((C-V)\) techniques. A plot of sheet Hall concentration \(n_s\) vs \(d\) gives accurate values of \((N_D - N_A)\) and \((w_i + w_e)\), the sum of the surface and interface free-carrier depletion widths, respectively. The \(C-V\) measurements verify the value of \(N_D - N_A\), and also give a good estimate of \(w_i\). By comparing the value of \(w_i\) with depletion theory, it is shown unambiguously that the interface depletion is mainly due to interface states, of concentration \(1.2 \times 10^{15} \text{ cm}^{-2}\) (below midgap). This result has important technological implications.

Hall effect measurements determine the *sheet* free-carrier concentration \(n_s\) in a semiconductor sample; thus, to get the *volume* concentration \(n\) it is necessary to know the sample thickness \(d\), i.e., \(n = n_s/d\). In thin samples, however, the effective electrical thickness \(d_{\text{eff}}\) can be significantly less than \(d\), because surfaces or interfaces can trap or immobilize some of the free carriers.\(^1,2\) In this letter we will consider the impact of these “depleted” carriers on Hall effect measurements in uniform, Si-doped, molecular beam epitaxial (MBE) GaAs layers grown on semi-insulating (SI) GaAs substrates. In such layers, a thickness \(w_i\) will be depleted due to surface states, and a thickness \(w_e\) due to a combination of interface states and substrate acceptor states. For these layers, \(n = N_D - N_A\) in the neutral regions, where \(N_D\) and \(N_A\) are the donor and acceptor concentrations, respectively. Then we have the simple relationship

\[
n_s = \int_0^w ndz = (N_D - N_A)d_{\text{eff}} = (N_D - N_A)(d - w_i - w_e) = (N_D - N_A)d - (N_D - N_A)(w_i + w_e),
\]

Equation (1) essentially defines \(d_{\text{eff}}\) and \((w_i + w_e)\) from a Hall effect point of view. We assume here that the mobility weighting of the Hall concentration is not important, which will be true unless the mobilities of the electrons in the layer vary strongly with depth. Further discussion of this point can be found in Ref. 4. For example, by using the formula presented there, along with assumed mobility variations in our layers of less than 10%, it can be shown that the mobility weighting effects on \(n_s\) are well under 1%.

Equation (1) demonstrates that if samples of varying metallurgical thickness \(d\), but identical in every other way, can be grown, then the slope of an \(n_s\) vs \(d\) plot will give \((N_D - N_A)\) and the intercept \((w_i + w_e)\). Fortunately, MBE growth processes allow precise control of \(d\) through the use of reflection high-energy electron diffraction (RHEED) oscillations, and also excellent control of other variables. Since the free-surface potential \(\phi_s\) is fairly well known,\(^3\) we can calculate \(w_i\) from this quantity, and then can calculate \(w_e\) from the sum \((w_i + w_e)\). Conversely, we can independently measure \(w_i\) with a capacitance-voltage \((C-V)\) experiment, and then calculate \(w_e\). From combined Hall effect and \(C-V\) data, along with a theoretical analysis of interface depletion effects, we will show unambiguously that the interface depletion in our case is due to a high concentration of interface states, and not to the expected filling of substrate acceptor states. This finding has important technological implications, and suggests further experimentation in initial growth conditions.

The MBE layers used in this study were of concentration \(1.69 \times 10^{17} \text{ cm}^{-3}\) [determined subsequently from Eq. (1)] and thicknesses 0.25, 0.50, 1.00, and 2.00 \(\mu\text{m}\). Note that the 0.25 \(\mu\text{m}\) layer is especially important because its doping and thickness are typical of material used for fabrication of metal-semiconductor field-effect transistors (MESFETs). The samples were grown in a Varian Gen II system directly (without buffers) onto four, 2 in., undoped, SI GaAs substrates, which were adjacent wafers taken from the same boule, in order to minimize substrate differences. The substrate temperature was accurately set at 580 °C by observing the oxide-desorption temperature, and variations during the runs were estimated at ± 3 °C. An As\(_4\) cracker was employed, and the RHEED pattern was 2 \times 4. The wafers were rotated during growth, and were grown one right after the other, with growth conditions held as constant as possible. RHEED oscillations, on a separate, stationary wafer, were used to set a precise growth rate \((0.7 \mu\text{m/h})\), and thus the thicknesses could be controlled to an estimated 0.01 \(\mu\text{m}\) over a small area. Because of possible thickness and carrier concentration variations at different points on a given wafer, the Hall samples (6 mm \times 6 mm) were each cut from
TABLE I. Sheet electron concentrations \( n_s \) and the "apparent" volume concentrations \( n = n_s/d \). The Hall \( r \) factor is 1.02 for \( n = 1.7 \times 10^{17} \text{ cm}^{-3} \) and \( N_A < N_D \).

<table>
<thead>
<tr>
<th>( d(\mu m) )</th>
<th>( n_s(10^{12} \text{ cm}^{-2}) )</th>
<th>( n(10^{13} \text{ cm}^{-2}) )</th>
<th>( n(10^{13} \text{ cm}^{-2}) ) corrected for Hall ( r ) factor</th>
</tr>
</thead>
<tbody>
<tr>
<td>0.25</td>
<td>1.443</td>
<td>0.577</td>
<td>0.588</td>
</tr>
<tr>
<td>0.50</td>
<td>5.595</td>
<td>1.12</td>
<td>1.14</td>
</tr>
<tr>
<td>1.00</td>
<td>13.80</td>
<td>1.38</td>
<td>1.41</td>
</tr>
<tr>
<td>2.00</td>
<td>30.50</td>
<td>1.52</td>
<td>1.55</td>
</tr>
<tr>
<td>true ( n )</td>
<td>1.66</td>
<td>1.69</td>
<td></td>
</tr>
</tbody>
</table>

The Hall effect results are shown in Table I. The slope of a least-squares fit of \( n_s \) vs \( d \) shown in Fig. 1 gives \( (N_D - N_A) = 1.66 \times 10^{17} \text{ cm}^{-3} \), without any Hall \( r \)-factor correction. For improved accuracy, we have calculated the \( r \) factor by a numerical solution of the Boltzmann transport equation. For low-compensation material, which was verified in our case by a 77 K mobility measurement, \( r = 1.02 \). For a higher compensation, \( N_A/N_D = 0.5 \), \( r = 1.06 \), but it is clear from these numbers that the \( r \) factor is not a major consideration at 296 K for carrier concentrations of \( 1-2 \times 10^{11} \text{ cm}^{-3} \). Thus, by using \( r = 1.02 \), the true carrier concentration is \( 1.69 \times 10^{17} \text{ cm}^{-3} \); then from the intercept of the slope in Fig. 1 we can calculate the total depletion thickness: \( (w_s + w_I) \approx 0.162 \mu m \). Theoretical values of depletion corrections suggested in the literature \(^1\), \(^3\) for \( n = 1.7 \times 10^{17} \text{ cm}^{-3} \) are close to this number; however, to our knowledge, this is the first time that \( (w_s + w_I) \) has been experimentally measured. Note from Table I the very large error incurred by not accounting for depletion in the calculation of \( n \); the 0.25 \( \mu m \) result is a factor 2.9 low, and even the 2 \( \mu m \) sample has an 8% error. Thus, for accurate MBE doping calibration, it is probably worthwhile to grow three or four samples of varying thickness, and apply a similar analysis.

We now compare with Hg probe \( C-V \) results, as shown in Fig. 2 for the 0.25 \( \mu m \) sample. It is well known that the \( C-V \) technique often gives inaccurate results, due to difficulties with forming the required Schottky barrier, measuring its effective area, accounting for series resistance effects, etc. However, in this case the carrier concentration plateau agrees well with the Hall effect value of \( (N_D - N_A) \), which we know to be accurate by virtue of Eq. (1) and Fig. 1. The tail region, of course, does not follow the \( (N_D - N_A) \) profile, which is known to be abrupt at \( d = 0.25 \mu m \), but is expected to follow the \( n \) profile reasonably well over the first decade of fall. Thus, from this tail region, we should be able to estimate \( w_I \) to within a Debye length (0.0104 \( \mu m \)) or so.

To be more quantitative, we write expressions for \( w_s \) and \( w_I \), in the depletion approximation:

\[
\begin{align*}
  w_s &= \left( \frac{2e(-\phi_s + \phi_{cb} - kT/e)}{e(N_D - N_A)_{\text{layer}}} \right)^{1/2} \\
  &= 0.0919 \left( -\phi_s + \phi_{cb} - \frac{kT}{e} \right)^{1/2} \mu m \\
  &= 0.0744 \mu m, \text{ if } \phi_s = -0.7 \text{ V} \\
  w_I &= \left( \frac{2e(-\phi_{sub} + \phi_{cb} - kT/e)}{e(N_D - N_A)_{\text{layer}} \left[ 1 + \left( (N_D - N_A)_{\text{layer}} / N_A - \text{sub} \right) \right]} \right)^{1/2}.
\end{align*}
\]
Here, the surface potential $\phi_s$ is usually given in the literature as approximately $-0.7$ V, the channel (neutral region) potential is calculated for this $n$ to be $\phi_n \approx -0.0191$ V, the thermal potential at 296 K is $kT/e \approx 0.0255$ V, the substrate potential $\phi_{ssub}$ is calculated as $-0.634$ V, the substrate acceptor concentration $N_{A,sub}$ is calculated from measurements of $N_{EL2}$ and $n_{sub}$ to be $3.45 \times 10^{15}$ cm$^{-3}$, and $N_{As, int}$ is the sheet acceptor density (below the Fermi level) at the layer/substrate interface. Equation (2a) is well known, whereas Eq. (3a) will be discussed more fully in a full length paper. Basically, Eq. (3a) simply represents the fact that $w_s$ is only 1.6 Debye lengths; a more accurate analysis will depend on a better theoretical understanding of the C-V curve itself.

Thus, we have unambiguously demonstrated that most of the interface depletion is due to interface states, not substrate acceptor states. Then, from the condition shown in Eq. (3c), we can calculate $N_{As, int} \approx 1.2 \times 10^{12}$ cm$^{-2}$. This is a technologically important finding, because it shows that almost 30% of the electrons in our 0.25 $\mu$m layer (which is a typical MESFET layer) are lost to interface states. Reduction of $N_{As, int}$ would lead to lower source resistance in a MESFET, an important consideration. In the future, we will be looking at the effects of different growth conditions on $N_{As, int}$.

A final remark concerns the potential use of the Hall effect experiment to study the effects on $\phi_s$ of various surface passivations, such as the recently investigated sulfide treatments. The Hall effect is especially well suited for such investigations, because the original sample surface is unperturbed by a Schottky barrier, or light irradiation, as is necessary in some of the other methods for studying $\phi_s$. Thus, a sample for which ($w_s + w_s$) has been determined, such as that in this study, can be subjected to a surface treatment and then remeasured, leading to an accurate value of $\Delta(w_s + w_s) = \Delta w_s$ (since $\Delta w_s = 0$); from $\Delta w_s$, it is possible to determine $\Delta \phi_s$. This technique is very simple and accurate and takes a minimum of time and effort. Results will be reported in the future.

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![Figure 2](image-url)